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A critical review on thermal conductivity enhancement of graphene-based nanofluids

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Highlights

- To determine the key parameters to develop highly-efficient heat transfer fluids
- Preparation of graphene-based nanofluids with different types of graphene
- Effect of the used graphene and the preparation approach on the nanofluid thermal conductivity
- Quantitative comparison of the enhancement of thermal conductivity of the graphene-based nanofluids

Keywords

Nanofluids; graphene; thermal conductivity; chemical modification;
performance comparison

Abstract

Nanofluids which consist of nanoparticles added to conventional fluids (or base fluids) are considered as promising heat transfer fluids. Compared to metal, metal oxide nanoparticles and carbon nanotubes, graphene with its extremely high intrinsic thermal conductivity became the best candidate to design nanofluids. Such nanofluids have the potential to be highly-efficient heat transfer fluid by reducing loss of heat and increasing cooling rates. Over the last ten years, graphene-based nanofluids have shown significant thermal conductivity enhancements, however due to the numerous and interlinked parameters to consider, optimisation of their efficiency is still challenging. The present review article analyses and discusses the reported thermal conductivity in term of performance with respect to the amount of the used graphene to develop the prepared nanofluids. The enhancement of thermal conductivity must meet the minimal graphene amount due to its production cost and because graphene nanoparticles induces high viscosity in the nanofluid leading to higher energy consumption for the heat transfer systems. Unprecedented in the literature, this work proposes a simple approach to quantitatively compare the enhancement of the thermal conductivity of the nanofluids. The thermal conductivity performance parameter introduced could be applied to all nanofluid families and may become a reference tool in the nanofluid community. Such tool will help to determine the optimal preparation conditions without compromising the superior thermal performances.

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Nomenclature

GE	Graphene
NP	Nanoparticle
TC	Thermal Conductivity
DIW	Deionized Water
EG	Ethylene Glycol
Wt %	Weight Percentage
SLG	Single Layer Graphene
GNS	Graphene Nanosheets
C/O	Carbon Oxygen ratio
DLG	Double Layered Graphene
TLG	Triple Layered Graphene
FLG	Few Layers Graphene
MLG	Multi Layered Graphene
GNF	Graphene based nanofluid
GNP	Graphene Nanoplatelets
GF	Graphene Flakes
GO	Graphene Oxide
rGO	Reduced Graphene Oxide
CVD	Chemical Vapor Deposition
HFGO	Highly Fluorinated GO
HRG	Highly Reduced Graphene Oxide
PVP	Polyvinyl Pyrrolidone
PG	Propylene Glycol
SDBS	Sodium Dodecyl Benzene Sulfonate
DOC	sodium deoxycholate
TPS	Transient Plate Source
THW	Transient Hot Wire
PEG	Polyethylene Glycol
TPS	Transient Plate Source
KPS	Potassium persulfate
LPE	Liquid phase exfoliation
AAD	Average Absolute Deviation (%)
GA	Gallic Acid
TEA	Triethanolamine
NMP	N-methylpyrrolidone

NDG	Nitrogen Doped Graphene
HFGO	Highly Fluorinated Graphene Oxide
SAS	sodium 4-aminoazobenzene-4-sulfonate
SDS	Sodium Dodecyl Sulfate
GA	Gallic Acid
T	Temperature
meas.	Measurements
conc.	Concentration
TC_{30}	Thermal conductivity enhancement value in % at 30°C
$TC_{nf}(30^{\circ}C)$	Thermal conductivity of the nanofluid at 30°C
$TC_{bf}(30^{\circ}C)$	Thermal conductivity of the base fluid at 30°C
P_{TC}	TC_{30} (%) divided by the concentration of graphene (wt.%) used to prepare the corresponding nanofluid.
C_{GF}	Concentration (in wt.%) of the employed GE to prepare the nanofluid

I. Introduction

Nanofluids, which consist of nanoparticles (NPs) added to conventional fluids (base fluids), have been considered as promising heat transfer fluids since their discovery by Choi and Eastman in 1995[1]. Many studies have shown the superior heat transfer properties of nanofluids, especially their thermal conductivity (TC) [2–9]. The role the nanofluids can play in thermal management is now widely recognized. The main idea is to disperse NPs possessing high thermal intrinsic conductivity in a solvent (or base fluid) in order to enhance the overall TC of the nanofluid. Therefore, the TC enhancement of the nanofluid is evaluated by taking the based fluid (with or without surfactant) TC as reference.

The nanofluids are envisaged to be used in cooling systems, for electronics or industrial engines requiring high thermal dissipation. And, efficiency of renewable energy like solar based on heat recovery and maximization of thermal transfer for storage or conversion would be boosted with the use of nanofluids[10,11].

Two main NP families have been investigated for nanofluids: metal based NPs (metals, oxides) and carbon nanomaterials (nanographite, nanodiamonds, carbon nanotubes and

graphenic species). The properties of the NPs commonly used for nanofluid preparation are given in Table 1. This Table shows that both thermal and electrical conductivities of some nanoparticle species can be really high compared to usual heat transfer fluids, evidencing the interest of nanofluids.

Table 1 Physical properties of the commonly used nanoparticles and base fluids to prepare nanofluids.

Nanoparticle	Density Kg.m^{-3}	Electrical conductivity (S.m^{-1})	Thermal conductivity ($\text{W.m}^{-1}.\text{K}^{-1}$)
Cu	8933 [12]	5.96×10^7 [12]	400 [12]
Co	8900 [12]	1.60×10^7 [12]	100 [12]
CuO	6310 [13]	10^{-3} [14]	33 [15]
TiO ₂	3900 [16]	10^{-1} [17]	5.6 [18]
Al ₂ O ₃	3500 - 3900 [16]	10^{-10} [19]	6.9 [20]
Fe ₂ O ₃	5180 [12]	0.112×10^6 [12]	80.4 [12]
Carbon nanotubes	~700 - 1700 [21]	$10^6 - 10^7$ [21]	~ 3000 [22]
Graphene	20 - 2000 [23]	$\sim 6 \times 10^8$ [23]	~ 5000 [23]
GO	10 - 1910 [24]	5.7×10^{-6} [23]	0.5–18 [23]
rGO	10 - 1910 [24]	$10^2 - 10^5$ [23]	1390–2275 [23]
water	1000	5.5×10^{-6} [25]	0.596 (20°C) [26]
ethylene glycol	1110	107 [25]	0.257 (20°C) [27]

The intrinsic TC of the metal based NPs is in the $6\text{-}400 \text{ Wm}^{-1}\text{K}^{-1}$ range which is of two/three orders of magnitude higher than the commonly used base fluids, [10] *e.g.* deionized water (DIW), ethylene glycol (EG) or mixtures of both of them. For example, Aberoumand *et al.* experimentally studied the TC of CuO nanoparticles (mean diameter 1 nm) in motor oil to investigate the effect of concentration and temperature on the TC of the nanofluids [28]. The concentrations of NPs in the base fluid were 0.2, 0.5 and 1 % by weight. An enhancement of 49 % in TC at 30 °C was found for 1 weight percentage (wt.%) of CuO. Fedele *et al.* studied the effects of adding titanium oxide nanoparticles, TiO₂ in water [29]. The results indicated that with 35 wt.% of TiO₂, the nanofluid TC was increased by 23 % at 30 °C compared to DIW alone. Other factors like the nanoparticles size can affect the TC of nanofluids. Sharifpur

et al. have investigated the effect caused by the addition of Al_2O_3 of different sizes (31 nm, 55 nm and 134 nm) to glycerol between 20 and 45°C [30]. The results showed that the maximum of improvement (19.5 % in relative TC at 30 °C) was reached for the smallest particles (31 nm) for the same NPs concentration (4 vol.%).

Other than metal or metal oxide NPs, carbon nanomaterials such as carbon nanotubes and graphene (GE), which combines lightness, chemical stability, high surface area, excellent mechanical properties and exceptional electrical and thermal properties, became a candidate of great interest to design high thermally conductive nanofluids [31]. GE which consists of carbons arranged in a regular sp^2 bonded atomic-scale honeycomb pattern possess a remarkable intrinsic TC. In addition to TC enhancement of nanofluids their stability is another crucial challenge. Aggregation of carbon based-NPs is usually observed due to lack of affinity towards the base fluid. Attractive forces between them lead to sedimentation and finally nanofluid destabilization which hinders their scale-up transfer. In the case of GE, compared to metal-based NPs, its lightness is particularly favorable to prevent sedimentation effects. However, pristine defect-free GE is highly hydrophobic and has high tendency to form aggregates in aqueous solvents. To solve this problem, great efforts have been devoted to improve the dispersion stability of pristine GE by various approaches such as mechanical agitation, ultrasonic treatment and their chemical modification of GE surface with surfactants (non-covalent functionalization) or introduction of functional groups (covalent functionalization) [32–35]. These two approaches for GE surface modification have each their pros and cons. The covalent functionalization involves the introduction of high number of oxygen-containing functional groups that inevitably damage the GE network, which negatively affects its physical properties including the TC [36]. The non-covalent approach consists in adding surfactants that adsorbs onto the GE's external surface [37] forming a coating layer known to be responsible for limiting heat transfer within the GE nanofluid

[38,39]. In both chemical approaches, significant TC enhancements mainly require high GE concentration which is prejudicial for developing low energy consumption systems due to the consequently increase in nanofluid viscosity [40–42]. More generally, because of the relatively high production cost of NPs, and GE is not an exception [43], the GE amount has to be minimized within nanofluids. The challenge for GE based nanofluids (GNFs) is indeed fully related to the ratio between TC enhancement and the amount of used GE to reach this enhancement.

Various sources of GE with different characteristics, available in large volumes from top-down methods [44,45] are used for nanofluid preparation. Among the most used graphene derivatives, graphene nanoplatelets or graphene flakes have a poorly defected structure and graphene oxide (GO) [46] bears a lot of oxygen-containing groups. GO can be reduced to produce reduced GO (rGO) by following straightforward procedures [47]. Consequently, the structure of graphene is partly restored as well as its properties.

There is a variety of GE based materials and different methods can be used for their functionalization and their use for nanofluids preparation. In general, the TC of GNF nanofluids depends or increases with temperature and it is usually improved as the amount of GE is increased within the nanofluid (in the case of good conductive GE and without sedimentation). GNF have been the topic of several review papers these last years. Among them, the preparation methods used to produce these nanofluids and the involved stability issues have been widely discussed [32–34]. The TC enhancement is as well of utmost importance for the GNF topic. In 2016, two review papers analyzed the thermal properties of GNFs [48,49]. Indeed, Rasheed *et al.* investigated the factors affecting TC and tribological properties of GNF such as the nanoparticles concentration, the temperature and the graphene size [48]. Besides, in the review by Sadeghinezhad *et al.* [49], different applications related to thermal properties of GNF have been analyzed by discussing the numerous parameters that

could affect the thermal efficiency. However, it remains difficult to directly compare the TC enhancement of the different nanofluids prepared. And despite the large number of GNF studies, the lack of understanding of the precise mechanisms and the relevant parameters affecting the GNFs properties has been pointed out. Identifying the optimum set of features that they must possess to maximize their properties is an open question. From there, any selection of the best GE type or the most promising functionalization approach cannot be done. Moreover, high presence of GE will increase the nanofluid viscosity triggering a higher energy consumption for heat transfer systems because higher pumping power is necessary [50,51]. This is the reason why minimizing GE concentration for nanofluid development is crucial [43]. In the present review, we propose a new and rigorous approach in order to provide a useful performance scale for GE based nanofluids. For that purpose, a performance parameter of GNF TC has been introduced. This new approach in the field could be applied to any other nanofluid type. Such performance parameter is of great usefulness for the nanofluid community and industry because such knowledge might become the basis to optimize GNF development and prepare high-performance GNF combining high stability and superior TC/GE amount ratio.

The present review is organized in three main parts. The first part gives a broad knowledge on the different types of GE nanomaterials used for nanofluid preparation, a focus is made on GE TC and the main methods for nanofluid preparation are briefly described. In the second part, an in-depth analysis of the TC performances is proposed. Finally, the third part will discuss the findings of the critical analysis and will give some recommendations for future works and development in GE nanofluids.

II. Graphene nanomaterials for nanofluids

II.1 Graphene properties

GE is known for being a nanomaterial holding outstanding thermal and electrical properties. Nowadays, GE possesses a wide domain of applications, from the aerospace industry to the microelectronics and the energy management sector. GE also serves as elemental base of diverse carbon allotropes, for example by folding 2D GE sheets in cylinders is possible to form 1D carbon nanotubes. Furthermore, graphite is obtained when several GE sheets are stacked together. The GE history can be tracked back as far as 1859 when Brodi *et al.* did the first attempt to synthesize it [52]. Nevertheless, GE's breakthrough really started in 2004 when Novoselov *et al.* succeeded to synthesize an isolated GE sheet [53].

GE's atomic lattice follows a honeycomb pattern with the thickness of a single carbon atom. Carbon atoms are bonded to three other carbon atoms by a strong double covalent bond while each atom in the lattice suffers from a sp^2 hybridization [54,55]. This hybridization is the origin of all GE's outstanding properties since the arrangement forms a perpendicular orbital to the sheet plane, inducing the behavior of a semiconductor or semimetal in the GE. The no existence of a band gap and the no overlapping of the GE's π bands (valence bands) and π^* bands (conduction bands) is another consequence of this feature [56,57]. Therefore, GE possesses the highest mobility of charge carriers among all known materials, with a value that reaches $200,000 \text{ cm}^2 \text{ V}^{-1} \text{ S}^{-1}$ [53].

In addition to remarkable electronical properties of GE, it also possess the highest TC ever reported in literature, with values of $5000 \text{ Wm}^{-1} \text{ K}^{-1}$ at room temperature [23,58,59]. The previously mentioned properties explain the growing interest of researchers in exploiting GE in composites as reinforce material.

II.2 Graphene types

Nowadays, the word “graphene” designs a wide group of GE derivative materials existing in the literature. Since their discovery, the demonstrated difficulty and cost of synthesizing a perfect single layer of graphene (SLG) or graphene nanosheet (GNS) has given rise to the existence of diverse GE derivative materials. The main difference between them is a variation on the number of layers, the lateral size and the presence of different functional groups on the GE surface.

The presence of defects is common in the GE derivative materials. These defects are introduced during the synthesis method because of the commonly harsh preparation conditions. Their existence is then difficult to avoid. However, these functional groups are also an important parameter to manage in such modified GE properties. The amount of functional groups attached on the GE surface can be defined by the carbon/oxygen ratio (C/O). This value impacts certain properties of GE. In other words, the diverse GE derivatives prepared possess different physical and chemical properties.

In the past years, a great effort has been made to homogenize the names given to the GE derivative materials [60]. Therefore, before performing a detail review of the GE based nanofluids, we first describe the most important graphenic derivative materials used for this purpose. The Figure 1 below represents the structural differences between some of the main existing GE derivatives. The axes represent the number of layers, the C/O ratio content and the GE average lateral size respectively.

Figure 1 Possible GE derivatives classification following the number of layers, average lateral dimension and carbon/oxygen (C/O) ratio located in the three different axes properly. Reproduced with permission from Wick et al. [61]

- The double layered (DLG) and triple layered graphene (TLG) refer to materials formed by two and three GE sheets, respectively. Both are included in a more general category called few-layer graphene (FLG). In fact, a FLG is defined as a 2D carbon material containing between 2 and 5 well-defined and stacked GE layers. FLG presents a high specific surface, hydrophobicity and excellent thermal and electric properties. Beyond 5 and until 10 stacked GE sheets, the material is called multi-layered graphene (MLG). The previously described GE derivatives lay over the marked “graphene nanosheet” presented in Figure 1.

- The graphene nanoplatelets or graphene flakes (GFs) are the graphenic derivative materials that, as it can be observed in Figure 1, contain more than 10 and until 30 GE honeycomb lattice sheets. These graphene nanoplatelets present a high hydrophobicity, which hinders their water dispersion. Even though their elevated number of GE layers, this material is still considered as a nanomaterial since the thickness is around 0.335 nm. For its remarkable electrical properties, GNPs are mainly used in electric applications like supercapacitors [62].
- The graphene oxide, well known as GO, originates from synthesizing a graphene derivative by an easy and cost-effective method. GO is mainly synthesized by oxidation and exfoliation hence this material is a chemical modified GE with a great content of oxygen type carboxylic (COOH), alcohol (-OH) and ether (-COC) functional groups attached on the basal plane. The presence of the oxygenated groups on GE surface makes the GO water soluble [63]. However, the existing functional groups disrupt the sp^2 hybridization bonding, distressing and lessening the GO thermal, mechanical and electrical properties. As shown in Figure 1, the nano GE oxide is characterized by a C/O ratio below 50 and a small lateral dimension.
- Later, reduced graphene oxide (rGO) was developed to improve the existing GO properties. Also, rGO is a particular GO that has been chemically treated to restore some of the damages caused by the oxidation in the sp^2 bonds while conserving GO water solubility [47]. The G/O ratio of the rGO is higher than that of GO, thus even if it is not represented, the rGO would be located further left than the GO in Figure 1. GO and rGO low production cost and high scalability make possible to be applicable in conductive inks, battery electrodes, sensors, polymer fillers [64].

Additionally to the previously mentioned GE derivatives, some carbon derivatives such as the GE quantum dots (GQD) are also available, GQD being a material showing quantum confinement and edge effects [65].

II.3 Graphene synthesis methods

It is clear that the different GE derivatives would be produced from different synthesis methods. In this case and as in all nanomaterials, the synthesis methods can be classified following two main approaches: bottom up and top down. The bottom-up approach consists of synthesizing the material from its very low parts (atom by atom) until the assemblage of the whole while the top-down approach involves the reduction, exfoliation or milling of a macroscopic material until the production of the nanomaterial.

Figure 2 summarizes both approaches as well as the most common techniques used to synthesize GE.

Figure 2 Schematic representation of the top-down and bottom-up approaches.

Reproduced with permission from ref. [66]

The Chemical Vapor Deposition (CVD) is the most representative process of the bottom-up approach. The CVD method is well recognized for allowing the successful production of pristine GE with a relatively well controlled number of layers (typically FLG, MLG and GNP) of several cm^2 wafer-scale on metallic nickel, copper or even cobalt substrates serving as GE growth catalysts [67]. The process consists in injecting a gaseous

mixture of a carbonaceous precursor often methane (CH_4), with hydrogen (H) into a furnace heated up to 800°C . The CH_4 molecules dissociate at the heated metal surface, then they recombine and segregate into GE sheets. GE synthesized by this method is of high quality and is particularly used for research and development in electronics. Otherwise, the main drawbacks are the high amount of metallic and carbonaceous impurities present in the outcome material [68].

The solvothermal synthesis, a less known method of the bottom-up approach, involves the synthesis of GE by a reaction between an alcohol, which is the carbonaceous precursor, and an alkali metal, mostly sodium. Alcohol and sodium are mixed under an inert atmosphere and placed in a reactor at 200°C . The obtained product is then pyrolyzed (i.e. heated to temperatures around 800°C inside an inert atmosphere). The resulting powder is then washed with hydrochloric acid to remove the impurities [69]. Following this washing, graphene in form of GO, FLG and MLG are obtained. Nevertheless, the final product yield is poor and the material suffers from a great amount of defects and impurities, mostly due the presence of sodium oxide (Na_2O).

Historically, graphite exfoliation is the first acknowledged technique to produce GE and it is an example of a top-down synthesis method. Besides, the exfoliation can be done either mechanically or chemically. In Figure 2, it is possible to distinguish both methods; in this case, the chemical exfoliation is denoted as “exfoliation” while mechanical exfoliation is denoted “mechanical exfoliation”.

GE synthesis method from mechanical exfoliation is based on the application of transverse or longitudinal stress on the graphite surface in order to prevail the van der Waals binding energy. Many agents can be used as stress appliers such as scotch tape [53] and electric field [70] making the mechanical exfoliation an efficient synthesise method. The

relevance of this method is due to the high quality of the obtained “pristine GE derivatives”. The GE derivatives are mostly FLG and GNF. However, the production yield is poor, making this technique not suitable for large-scale production. Nonetheless, a novel approach associated with the use of ultrasonication of GE in liquids has achieved a high production yield with low defects in the produced FLG [71,72] .

In the case of chemical exfoliation, harsh conditions are used to overcome the attractive van der Waals forces between the graphite layers. The solvents are either toxic organic solvents such as cyclohexanone (CYN), dimethyl sulfoxide (DMSO) or water. Use of water is particularly difficult since pristine GE is highly hydrophobic. Consequently, exfoliation in water is achieved with the help of surfactants that facilitate the exfoliation of graphite and stabilize (pre-disperse) the GE [73,74]. For instance Bourlinos *et al.* used pyridine as well as some non-aromatic solvents like ethyl acetate, vinyl acetate, methyl chloroacetate, among others to extract GE from graphitic powder, obtaining dispersed graphite at variable concentrations and different yields according to the employed surfactant [75]. The predilected synthesized product of the chemical exfoliation is the exfoliated GO as it is indicated in Figure 2.

The first chemical exfoliation method was performed by Brodi *et al.* [52]. This method involved the use of strong acids such as nitric acid (HNO_3), representing a high risk and danger compared to the low production yield [52]. Years later, Staudenmaier *et al.* and Hofmann *et al.* improved Brodi’s method achieving an increase in the produced GO yield [76–78]. Nevertheless, the production method was still being dangerous by the utilization of strong acids. Afterwards, Hummers *et al.* developed a chemical exfoliation procedure that currently is the most common method to synthesize chemically exfoliated GE [46]. This approach employs strong oxidants such as potassium permanganate (KMnO_4), hydrogen peroxide (H_2O_2) and concentrated sulfuric acid (H_2SO_4). The compounds oxidize the GE planes and

this oxidation reaction causes the distance between the GE sheets to increase. Both washing and decantation cycles must be carried out in order to return to a neutral pH, then the resulting GO should be dried. Hummers' method allows the production of GO in relatively large quantities with a limited presence of defects. In the recent years, modifications of this procedure have been reported in the literature, known as the "modified Hummers' methods" [79,80].

The rGO can be obtained by chemical reduction of GO [81,82]. This reduction can be simply achieved by adding different reducers to a dispersion of GO in water such as ascorbic acid, hydrazine and sodium borohydride. The final product, the highly rGO (HGR), most commonly denoted as rGO, possesses a small amount of oxygenated groups as it is evidenced in Figure 2. The fact that oxygenated groups are chemically reduced induces a partial restoration of the sp^2 network [83]. The reduction of these groups brings an increase in the thermal and electrical conductivity of rGO with respect to GO [81,82,84,85]. GO is an electrical insulator with a conductivity less than $10 \mu\text{S/m}$ (Table 1). Graphene conductivity depends indeed on the extend of oxygen groups [86]. By reducing them by a chemical or thermal method, an increase in electrical conductivity was then observed in the 1-2000 S/cm range depending on the structural quality of the studied rGO [85,87,88]. Moreover, an increase of thermal conductivity of composites and films when GO was reduced to rGO was widely reported [87,89–92]. Such an enhancement is attributed to the increase in thermal conductivity of rGO compared to that of GO as evidenced in Table 1.

II.4 Thermal conductivity of graphene

Even if the TC measurement depends on the type of the investigated method (theoretical or experimental), the reported values for GE being in the $2000 - 10\,000\text{ Wm}^{-1}\text{K}^{-1}$ are remarkably high [58,59,93]. These studies are mainly focused on CVD synthesized GE. Topological defects are often introduced in the outcome material since during the CH_4 dissociation process the GE structure can suffer from regional overlapping and rupture [94]. Additionally, most of research on TC has been carried out in suspended CVD grown GE. Experimentally, TC is usually measured by the optothermal Raman technique but the presence of a substrate leads to uncontrolled effects due to phonon leakage across the GE-substrate interface [95]. TC of GE may be affected by several factors including the lateral dimensions of the layer, the edge states, the number of layers and the number of defects [96–99]. The majority of studies, either experimental or theoretical, have reported that TC is reduced by increasing defect density within the GE structure by functionalization [82,100–103] or doping [36] (Figure 3). It was also claimed that depending on the functionalization level in the case of grafting of nitrophenyl groups covalently bonded to GE, TC could be even increased [104]. The intrinsic TC of GNP, GF, GO or rGO has been poorly investigated so far.

Chemical modifications of the GE are an alternative to improve its properties especially its high hydrophobicity. The chemical modification of GE surface can be done either from covalent or non-covalent approach. Both types of functionalization are commonly used, at an equivalent level, to design GNFs. In the following, the influence of these two different types of functionalization on GE TC is evaluated.

Concerning the covalent functionalization, the introduction of defects at the functionalized carbon site unavoidably modifies the sp^2 hybridization leading towards a sp^3 hybridization since the $\text{C}=\text{C}$ double bonds must be “broken” to serve as attachment sites for the foreign molecules. The additional phonon scattering due to the defected network reduces the TC of the modified GE, as illustrated in Figure 3.

Figure 3 Dependence of the TC on the density of defects of SLG. The experimental data are shown by squares, circles and triangles. The solid curves are calculated curves. p is a parameter called the specularity parameter; for $p = 0$ the scattering is fully diffuse and for $p = 1$, there is no extra thermal resistance due to diffusion. Reproduced with permission from ref. [36]

Regarding the non-covalent approach, for which the sp^2 network remains intact, negative effects on the TC due to addition of surfactants have been reported [38,39]. The surfactants are usually organic compounds having intrinsic low TC. Even if low concentration are usually used, their presence in the base fluid is known to diminish the TC of the solvent-surfactant mixtures compared to the solvent alone [105]. Moreover, it has been already

reported that the surfactant molecules on the GE surface acts as an insulating layer, limiting the transfer between the dispersed GE and consequently decreasing the thermal and electrical conductivity of the resulting nanofluid [39,106]. Yu *et al.* reported that 5 wt.% of polyvinyl pyrrolidone (PVP) used as surfactant decreased the TC of EG-PVP mixture by about 2 % compared to that of pure EG [107]. In another work, an ionic surfactant, sodium dodecylbenzene sulfonate (SDBS), used at a concentration of 1 vol.% reduced the TC of Al_2O_3 -based nanofluids. This effect has been shown to be even worse by increasing the SDBS/ Al_2O_3 volumetric ratio [108].

III. Graphene based-nanofluids

Generally, nanofluids can be prepared by either a one- or two-stage approach. Mostly DIW, EG, propylene glycol (PG), ionic liquids and oils, like kerosene, are used as a base fluid. Also, a mixture of the previously mentioned solvents can be use as based fluids. The one-step method for preparing nanofluids is less used than the two-steps approach. It involves the synthesis of the NPs and their simultaneous dispersion in the base fluid using different chemical or physical methods such as physical/chemical vapor deposition or laser ablation. For example, Akoh *et al.* developed a one-step synthesis method via evaporation/condensation [109]. A metal vapor is directly condensed into nanoparticles when it comes into contact with a low vapor pressure fluid, in this case EG. The resulting nanofluid is formed by metallic nanoparticles with an average diameter of 2.5 nm, suspended in the EG solution . Laser ablation is a recent technique that has proven its effectiveness. An example is the work of Mortazavi *et al.*, whom achieved the synthesis of a graphene material with thickness of 0.07 nm inside an liquid nitrogen atmosphere [110].

The one-step technique has the advantage of improving the stability with less NP agglomeration in the resulting GE-based nanofluid. Although, its two main disadvantages are

the presence of residual reactants in the outcome solution due quite often incomplete chemical reaction and the elevated production cost that makes it unsuitable for large-scale production. Besides, due to its complexity, the one-step approach is rarely used to produce GNF nanofluids.

Unlike the one-step method, the two-step method is more suitable for high-scale production. Moreover, this is the most economical and widely used method for preparing nanofluids, including GNFs. Here, the nanoparticles are first synthesized in powder form, followed or not by chemical treatments (covalent functionalization). Then, the desired amount of solid is added to the host fluid (DIW, EG, etc.). Additives such as surfactants can be added to the solution to improve the nanoparticle dispersion by non-covalent functionalization. Most of the times, the solution must be agitated by ultrasonication to promote the solute dispersion of NPs, homogenization [32,111] and reduce GE aggregation.

A homogeneously well dispersed GE is essential to obtain long-term thermal properties of nanofluids [112]. Stability is recognized as a long-standing problem. Nanofluid destabilization limits the development and use of GNFs. The commonly observed weak dispersion stability of GE in the base fluid is a direct consequence of its high hydrophobicity and its resulting tendency to form aggregates.

For a better understanding of the nanofluids stability, it is important to know the factors that affect the NP dispersion. The dispersion stability depends on the interactions of various microscopic forces that are exerted on the NPs. Truth is that differences in the solution density causes the gravity forces to inevitable sediment the dispersions. Meanwhile, NPs in the fluid undergo a random thermal movement, the "Brownian motion" which is defined as an irregular and random motion of a large particle immersed in a fluid and which is subject to shock interactions with the small molecules of the fluid [113]. If the repulsive forces

between these molecules or NPs is not strong enough, attractive (often van der Waals) interactions will induce their aggregation and nanofluid destabilization.

It is important to mention that, in recent years, a great effort has been devoted to overcome this problem. Chemical modifications of carbon nanomaterials and especially GE via non-covalent or covalent functionalization is commonly used to overcome its high hydrophobicity and enhance its dispersion. The functionalization modifies GE surface and therefore, the carbonaceous materials would be able to counterbalance the van der Waals aggregation force.

The covalent functionalization involves grafting hydrophilic functions on the GE sheets, mainly oxygenated functions, by covalent bonds. The main reactants in this process are acids and strong solvents. The remaining hydrophilic molecules on GE later improves its dispersion in the base fluid by electrostatic and steric repulsions [114–116]. Furthermore, while the pristine GE is chemically inert and zero band gap material, functionalized GE can be used for a new generation of engineered devices and the functionalization could even be used to tune the material's properties. Indeed, using defected GNS leads to better interface interactions and transfer phenomena between GE and the surrounding medium inducing a series of enhancement in some of the mixture properties such as mechanical, electrical and thermal compared to the pure surrounding medium (polymer, solvent...).

The non-covalent functionalization uses surfactants, molecules with a hydrophilic and a hydrophobic part, to disperse the GE in the nanofluid. The surfactants help to counterbalance the aggregation force since their hydrophobic part adsorbs on the GE surface while the hydrophilic part is in contact with the solvent, increasing the affinity of GE nanolayers with the base fluid by steric repulsion. Depending on the surfactant polar group nature (ionic, nonionic, cationic) electrostatic repulsions can also be exerted between the functionalized GE sheets. The pH also influences the repulsion exertion forces between the

surrounding fluid and the grafted compound [117]. The non-covalent functionalization is relatively simple to implement and allows to preserve the GE physical properties since it does not induce any damage on the sp^2 network.

Among the many existing techniques reported in the literature to measure the nanofluids stability, Zeta potential measurements, UV-Vis spectrometer and sedimentation photograph capturing figure are among the most common. The first two techniques offer a quantitative and trustful analyze while the third one is merely qualitative. Concerning the TC, among the diverse existing techniques, the Transient Plate Source (TPS) and the Transient Hot Wire (THW) methods are mainly used, as shown later in Table 2, due to their reliability and accuracy.

IV. Comparative analysis of thermal conductivity performances of graphene-based nanofluids

The literature concerning the thermal performance of GE-based nanofluids is extensive. Among the reported results, different sets of experimental conditions have given different results in TC enhancements. These experimental conditions are numerous: TC measurement device, temperature range investigated, GE amount, based fluid nature, GE type and chemical modification and presence of surface agents among others. Such complexity explains the difficulty to directly compare the reported TC enhancements found for different nanofluids. In this section, we propose a tool whose main goal is to serve as a performance indicator of TC enhancement.

For that purpose, this review proposes a rational comparison of the thermal conductivity of GNFs reported in the literature over the last 10-year of publications. This

review excludes the hybrid nanofluids. Table 2 gathers information extracted from the works published on GNFs and their TC enhancements to date. More than 60 papers and studies have been analyzed. For each analyzed work, the type of GE (and its source) used for the nanofluid development and the concentration range investigated are given as well as the chemical modification possibly applied to optimize the nanofluid preparation. This includes pristine GE nanomaterials (without functionalization) and GE having undergone either covalent or non-covalent functionalization. Regarding the nanofluid preparation, the used method and the chosen base fluid are also given. For the TC measurements carried out, the used method and the temperature range investigated are also specified. This detailed description of the nanofluid preparation and studied parameters is useful to compare the GE based nanofluid or to reproduce the reported experiments. With the aim of proposing a simple approach to quantitatively compare the enhancement of the measured TC of the nanofluid compared to the TC of the base fluid (without GE), the best TC enhancement at the same temperature was extracted from each study. Therefore, 30°C was selected as it is the most representative temperature investigated. This TC enhancement value in % at 30°C is referred to as $TCE_{30}(\%)$ and is defined by the equation (1).

$$TCE_{30}(\%) = \frac{TC_{nf}(30^{\circ}C)}{TC_{bf}(30^{\circ}C)} \times 100 \quad (1)$$

Where, $TC_{nf}(30^{\circ}C)$ and $TC_{bf}(30^{\circ}C)$ are the TC of the nanofluid and the base fluid at 30°C, respectively.

Beyond this TC enhancement value alone, it appears particularly relevant for nanofluid development to relate this enhancement to the amount of GE inducing the nanofluid TC increment. Interestingly, by simply dividing the $TCE_{30}(\%)$ by the concentration of GE (wt.%) used to prepare the corresponding nanofluid, we introduce a universal parameter useful to compare the performance between different nanofluids. This parameter, defined by equation

(2), is a thermal conductivity performance parameter (P_{TC}) directly linked to the efficiency of the used GE to enhance the TC of the prepared nanofluid. Therefore, this parameter is defined by equation (2).

$$P_{TC} = \frac{TC_{e30}(\%)}{C_{GE}(wt.\%)} \quad (2)$$

where C_{GE} is the concentration in (wt.%) of the employed GE.

P_{TC} corresponds to the TC enhancement induced by 1 g of the corresponding GE type dispersed in the base fluid for 100 g of GNFs. For example, if high GE concentration induces a high TC enhancement in the nanofluid, the P_{TC} value could be even lower than for a nanofluid for which the TC enhancement is less but prepared with a much lower GE amount.

As previously mentioned, high GE concentration in nanofluids leads to an increase in their viscosity. This is the reason why using low GE concentration is crucial. Furthermore, apart from TC enhancement, viscosity and more generally rheological behavior of nanofluids (not studied here) is another important parameter to study for the designed GNFs [41]. The used GE amount is an important parameter to consider for evaluating the nanofluid TC performance. And an optimal GE amount has to be found for nanofluid designing since it can represent around 80 % of the nanofluid production cost [43].

From the data gathered in Table 2, an analysis of the evolution of the GE nanofluid performances regarding their TC is proposed below.

Table 2 Graphene-based nanofluid preparation conditions, including GE source, functionalization approach, base fluid, GE concentration range and dispersion method, and thermal conductivity at 30°C along with the measurement method, the temperature range and the reported thermal conductivity at 30°C, TC_{30} and the calculated nanofluid TC performance parameter P_{TC} .

GE type		Nanofluid preparation						NF thermal conductivity					Published work	
								TC measurement		TC performance				
type	source	metho d	base fluid	functionalization		conc. range (wt.%)	method	method	T range (°C)	GE conc. (wt.%)	TCe ₃₀ (%)	$\mathcal{P}_{TC}$ (%/wt.%)	date	Ref.
				covalent	non- covalent									
GNP	commercial	2-step	kerosene	NO	oleylamine	0.0005 - 0.2	Ultrasonic probe, 40 min – 3h	THW (KD2 Pro) accuracy ±5%	20 - 70	0.2	23	115	2016	[118]
GF	commercial	2-step	DIW	NO	SDBS	0.093-0.28	Sonication 30 min	THW	10 - 50	0.28	35	126	2016	[119]
GNP	commercial	2-step	DIW	PEG	NO	0.01 - 0.1	Ultrasonic bath 200 W, 60 kHz, 2h	THW (KD2 Pro Decagon Devices, Washington,USA) average of 5 meas.	20 - 60	0.1	11	110	2019	[120]
GO	Hummers' method	2-step	EG	H ₂ SO ₄ :HNO ₃ oxidized	NO		Ultrasonication 30Hz, 30-45 min	THW (KD2 Pro Decagon, Canada) uncertainty < 4%	25 - 50	0.1036	18	174	2010	[116]
GO	Hummers' method	2-step	DIW	hydrogen exfoliated + H ₂ SO ₄ :HNO ₃ oxidized	NO	0.01 - 0.1	Ultrasonication during 30-45min	THW (KD2 Pro Decagon, Canada)	25 - 50	0.1	22	212	2011	[115]
GNP	commercial	2-step	DIW + EG (90:10)	sulfonic acid	NO	0.1 - 0.5	Ultrasonic bath 200W, 40 kHz	TPS (Hot Disk AB, Sweden) input time 4sec, accuracy 5%.	20-70	0.5	5	10	2017	[121]
GNP	commercial	2-step	Heavy duty diesel engine oil CH-4 20W-50	NO	Lipophilic polymer WinSperse 6020	0.5-3	Magnetic Stirring and ultrasonic bath	TPS (Hot Disk AB, Sweden)	20-70	3	23	7.7	2020	[122]

GNP	commercial	2-step	hydrogen ated oil	NO	NO	0.0025 - 0.01	Ultrasonic bath 3 h, 320W, 40 kHz	THW (KD2 Pro Decagon)	30-50	0.01	10	1000	2017	[123]
GO	commercial	2-step	DIW	NO	NO	0.001-0.5	Ultrasonication 5 min	TPS (TPS2500, Hot Disk Inc., Sweden) accuracy ±3%	30-80	0.1	19	190	2017	[124]
rGO	commercial	2-step	DIW	NO	PVP	N.A.	Ultrasonication 5 min.	TPS (TPS2500, Hot Disk Inc., Sweden) accuracy ±3%	30-80	0.02	9	450	2017	[125]
GO	Hummers' method	2-step	DIW	NO	NO	0.01 - 0.1	Sonication 130W, 42 kHz, 1h	THW (KD2 Pro Decagon Devices, Inc., Pullman, WA, USA) uncertainty 5%	25 - 40	0.1	22	220	2017	[126]
GNP	commercial	2-step	DIW + EG (1:1)	NO	NO	0.05 - 0.15	Ultrasonication 1h	THW (KD2 Pro Decagon Devices, Inc.) accuracy ±3%	(-20) - 50	0.15	16	107	2018	[127]
FLG	CVD	2-step	DIW	alkaline oxidation by KPS	NO	0.05 - 0.1	15 minutes sonication	THW (KD2Pro Decagon Devices, Inc.) accuracy ±0.001	10 - 60	0.075	51	680	2014	[128]
FLG	CVD	2-step	DIW	alkaline oxidation by KPS	NO	0.01 - 0.05	Ultrasonic water, 1 hour	THW (KD2 Pro device Decagon Devices, Inc.)	10 - 50	0.05	15	300	2013	[129]
GQD	commercial	2-step	DIW	NO	NO	0.0025 - 0.02	Ultrasonic Probe.	THW (KD2 Pro device Decagon Devices, Inc.)	20 - 60	0.02	9	450	2017	[130]
rGO	Hummers' method	2-step	DIW	NO	NO	0.1 – 0.37	N.A.	THW, error < 1.2%, average of 8 meas.	25 - 65	0.37	11	30	2011	[131]
GO	Hummers' method	2-step	EG	NO	NO	0.001 - 0.07	Probe ultrasonication 50 W, 15 min	THW (KD2 Pro Decagon Devises, Inc.)	20 - 70	0.07	31	443	2014	[132]
GO	Hummers' method	2-step	DIW	NO	NO	0.05 - 0.25	Ultrasonic Bath with 200 W at 60 kHz	THW, KD2 Pro (Decagon devices, Inc., USA) accuracy < 5 %,	10 - 40	0.25	45	180	2014	[133]

FLG	LPE (Liquid Phase Exfoliation)	2-step	PG:DIW (40:60)	NO	P-123 Pluronic	0.05 - 0.5	Probe sonicator, 125W pulse mode, 2s ON and 1 OFF	THW (THW-L2 device, Thermtest Inc., Richibucto Road, NB, Canada), average absolute deviation (AAD) 1%, average of 6 meas.	10 - 50	0.5	27	54	2020	[74]
GO	Hummers' method	2-step	DIW:EG (60:40)	NO	NO	0.01 - 0.10	Sonication bath, 40 kHz, 280 W and stirring 10 min	THW (KD2 Pro Decagon device) accuracy ± 5 , average of 6 meas.	25 - 45	0.1	13	130	2015	[134]
GNP	commercial	2-step	DIW	NO	NO	0.05 - 0.1	Ultrasonication probe	THW (KD2 Pro Decagon device, USA)	20 - 60	0.1	30	300	2016	[135]
GNP	commercial	2-step	DIW	NO	NO	0.025 - 0.1	Ultrasonication Probe	THW (KD2 Pro Decagon device USA), precision $\pm 0.1^\circ\text{C}$	15 - 70	0.1	27	270	2017	[136]
GO	commercial	2-step	DIW	NO	NO	0.001 - 0.045	Ultrasonication probe 350 W, 20 kHz, 45 min	THW (KD2 Pro Decagon devices Inc., US) error $< \pm 5\%$ and random error $< 1.2\%$	25 - 50	0.045	5	111	2017	[137]
rGO	Hummers' method	2-step	DIW:EG (70:30)	H ₂ , 200°C	NO	0.076 - 0.73	Intense ultrasonication 45 min	THW, uncertainty $< 61\%$	10 - 70	0.73	14	19	2013	[138]
GNP	commercial	2-step	EG	NO	NO	0.93 – 7.4	Ultrasonication 500 W, 25 kH and probe-type 750 W, 20 kHz	THW (LAMBDA, F5 Technologie, Germany)	10 - 90	7.4	29	3.9	2014	[139]
rGO	Hummers' method	2-step	ionic liquid: [HMIM]BF ₄	NO	NO	0.03 & 0.06	Ultrasonic probe, 25 W, 40 kHz, 8h	TPS (Hot Disk TPS 2500S, Hot Disk AB, Sweden) accuracy $< 3\%$	25 - 200	0.06	16	267	2014	[140]
FLG	commercial	2-step	ionic liquid: [HMIM]BF ₄	NH ₄ BF ₄	NO	0.01 - 0.05	Middly stirring (without sonication)	THP (TPS2500, Hot Disk, Sweden)	30 - 150	0.05	8	160	2017	[141]
rGO	Hummers' method	2-step	silicon oil	3-glycidoxypopyltrimethoxysilane (KH-570) + hydrazine	NO	0.01 - 0.07	6h in a ultrasonic bath	THW, average of 3 meas.	20 - 60	0.07	7	100	2013	[142]
NDG	Hydrothermal	2-step	DIW	NO	Triton X-100	0.01 - 0.06	Ultrasonication probe 60 min	N.A.	15 - 40	0.06	33	550	2016	[143]

GNP	commercial	2-step	DIW	NO	NO	0.025 - 0.1	Ultrasonication probe	THW (KD2 Pro Decagon Devices, Inc., Pullman, WA, USA)	15 - 40	0.1	27	270	2015	[144]
GNP	commercial	2-step	DIW	NO	NO	0.025 - 0.1	Ultrasonication probe 1200 W, 20 kHz	THW (KD2 Pro Decagon devices) accuracy $\pm$ 5%	15 - 40	0.1	27	270	2015	[145]
rGO	Hummers' method	2-step	DIW	polyphenol from red wine	NO	1.9 – 7.4	sonication bath (unspecified time)	THW (KD2 Pro Decagon Devices, USA) accuracy 5%	15 - 40	7.4	18	2.4	2016	[146]
NDG	Hydrothermal	2-step	DIW	NO	Triton X-100	0.01 - 0.06	Ultrasonication probe 1200 W, 20 kHz, 60 min	THW (KD2 Pro Decagon Devices, USA) accuracy < 5 %	15 - 40	0.06	35	584	2014	[147]
GNP	commercial	2-step	DIW	NO	NO	0.025 - 0.1	Ultrasonication probe 1200 W output power and a 20 kHz.	THW, (KD2 Pro Decagon Devices, USA) accuracy 5%, average of 10 meas.	15 - 40	0.1	27	270	2014	[148]
nano porous GE	CVD	2-step	DIW	NO	Ter-polymer	0.025 - 0.1	Ultrasonic	THW (KD2 Pro Decagon Devices)	15 - 40	0.1	4	40	2016	[149]
GO	Hummers' method	2-step	DIW:EG (70:30)	NO	SDS	0.1-0.3	Ultrasonic probe, 8h	THW (KD2 Pro Decagon Devices) accuracy $\pm$ 5%	30-70	0.3	12.6	42	2020	[150]
GNP	commercial	2-step	DIW	NO	NO	0.025 - 0.1	Ultrasonication probe 1200 W at 20 KHz	THW (KD2 Pro Decagon Devices, USA), error < 5%	15 - 40	0.1	25	250	2016	[151]
GNP	commercial	2-step	DIW	SAS	NO	3.7-9.25	Ultrasonication probe, 5 min, 2 h in ice-water bath	THW (KD2 Pro Decagon Devices, USA) accuracy < %5	15 - 40	9.25	16	1.7	2020	[152]
GNP	commercial	2-step	DIW	NO	NO	0.025-0.1	Ultrasonication probe, 750 W, 20 KHz	THW (KD2 Pro Decagon Devices, USA) accuracy 2–4%	15 - 40	0.1	21	210	2015	[153]
GNP	commercial	2-step	DIW	GA	NO	0.093	Ultrasonication probe for 20 min	THW (KD2 Pro Decagon Devices, USA) accuracy 5%, average of 20 meas.	20 - 45	0.093	16	173	2017	[154]
GNP	commercial	2-step	DIW	GA	NO	0.025 - 0.1	Ultrasonicated for 15 min with sonication probe	THW (KD2 PRO Decagon Devices, USA), accuracy 5%, average of 20 meas.	20 - 45	0.1	17	170	2017	[155]
GNP	commercial	2-step	DIW	eugenol, eugenyl acetate and b-caryophyllene (clove-treated)	NO	0.025 - 0.1	Ultrasonication for 10 min	THW (KD2 Pro device) accuracy $\pm$ 5%	20 - 45	0.1	15	150	2018	[156]

GNP	commercial	2-step	DIW	TEA	NO	0.025 - 0.1	Ultrasonication, 450 W pulsed mode 2 S on/off	THW (KD2 Pro Decagon devices, Inc., USA), accuracy < 5%, accuracy 0.1°C, 8 meas.	20 - 40	0.1	19	190	2016	[38]
GNP	commercial	2-step	DIW	NO	Gum Arabic	0.1	Ultrasonication probe, 750 W, 20 kHz, 15, 30, 60, 90 and 120 min.	THW, (KD2 Pro Decagon devices, Inc., USA) accuracy < 5%, average of 8 meas.	20 - 40	0.1	10	100	2016	[106]
GQD	CVD	2-step	DIW	NO	NO	0.05 - 0.5	Ultrasonic bath for 15 min	THW (KD2 Pro Decagon Devices Inc.; USA) error ± 0.05 W/mK	10 - 50	0.5	34	68	2019	[157]
GNP	commercial	2-step	DIW:EG (70:30)	NO	DOC	0.19 – 0.93	Ultrasonication probe, 700W 20 kHz during 2 h	THW (KD2 Pro Decagon Devices Inc, USA), uncertainty ±5.0 % (0.2 - 2 W/m K)	30 - 50	0.93	20	21.6	2017	[158]
GNP	commercial	2-step	DIW:EG (70:30)	NO	DOC	0.002 – 0.8	Ultrasonication probe, 700W 20 kHz, 2 h	THW (KD2 Pro Decagon Devices, USA), uncertainty ±5.0 % (0.2 - 2 Wm-1K-1), average of 10 meas.	30 - 50	0.8325	17	20.4	2017	[159]
GNP	commercial	2-step	DIW	NO	DOC	0.002 – 0.8	Ultrasoication probe 2 h	THW (KD2 Pro Decagon Devices, Inc, USA) uncertainty < ±5.0% (0.2 - 2 W/m K)	30 - 50	0.925	27	29	2016	[160]
GNP	commercial	2-step	DIW	oxidation with H ₂ O ₂ and treatment with xylitol and citric acid	NO	0.1	Stirring 30 min, RT. Sonication 1 h	THW (KD2 Pro Decagon devices, USA), error < ±0.9%	20 - 60	0.1	16	160	2018	[161]
HFG O	from fluorinated graphite polymer	2-step	NMP	NO	NO	0.01 - 0.10	sonication probe 6h	THW (KD2 Pro Decagon Device). 6 meas.	20 - 40	0.1	18	180	2014	[162]
GO	commercial	2-step	DIW	NO	NO	0.001 - 0.045	Ultrasonication probe 350 W, 20 kHz, 45 min	THW (KD2 Pro Decagon devices Inc., US)	25 - 50	0.045	5	111	2016	[163]
GNP	commercial	2-step	DIW	NO	NO	0.00025 - 0.005	Ultrasonication probe, 665 W, 20Khz, 1 h	THW (KD2Pro Decagon devices Inc., US)	25 - 50	0.005	6	1200	2016	[164]
GNP	commercial e	2-step	Havoline [®] XLC:water (50:50)	polycarboxylate chemically modified	SDBS	0.25 - 1	Stirring & ultrasonication 200 W, 20 kHz, 240 min	THW (KD2 Pro Decagon Devices, USA), uncertainty < 3%	20 - 70	1	6.6	6.6	2019	[114]

GO	Hummers' method	2-step	DIW	NO	NO	0.005 - 0.02	Ultrasonicated during 30 min.	THW (KD2 Pro Decagon, Canada)	20 - 70	0.02	17	830	2016	[165]
rGO	Hummers' method	2-step	ionic liquid: ([HMIM] BF ₄)	NO	NO	0.03 & 0.06	Ultrasonication 100 W, 40 kHz, 8 h & ultrasonication probe 25 W, 2 min	TPS (Hot Disk TPS 2500 S, Hot Disk AB, Gothenburg, Sweden)	25 - 200	0.06	16	267	2012	[166]
GO	Hummers' method	2-step	DIW	NO	NO	0.5 - 1.5	Stirring 1h & ultrasonication 60 min	THW (KD2 Pro Decagon), error < 1%	20 - 60	1.5	29	19	2019	[167]
GNP	commercial	2-step	DIW	oxidation with HNO ₃ and H ₂ SO ₄ at 1:3 ratio	NO	0.02 - 0.1	Ultrasonic bath 3 h.	THW (KD2 pro Decagon, USA), 16 meas. 4 h	20 - 40	0.1	12	120	2016	[168]
GO	Hummers' method	2-step	liquid paraffin	NO	NO	1.9 - 9.3	Stirring & sonication, 40 kHz, 150 W, 3 h	THW, accuracy < 1%	10 - 60	9.3	76	8.2	2010	[169]
GO	Hummers' method	2-step	EG	NO	NO	1.9 - 9.3	Stirring & sonication 40 kHz, 150 W	THW, accuracy < ±1%	10 - 60	9.3	58	6.3	2010	[170]
rGO	Hummers' method	2-step	EG	NO	SDBS	1.9 - 9.3	Stirring & sonicating 40 kHz, 150 W	THW. A transient short hot-wire (SHW) accuracy 1%	10 - 60	9.3	90	9.7	2011	[171]
rGO	Hummers' method	2-step	DIW	NO	NO	0.02 - 0.1	Ultrasonication probe	THW	20 - 60	0.1	17	170	2017	[172]
GNP	Commercial	2-step	DIW	NO	NO	0.0025 - 0.1	Ultrasonication probe 120 W, 40 kHz, 2h	THW (KD2 Pro Decagon), accuracy < ±5%	15 - 40	0.1	16	160	2018	[173]

NH₄BF₄ : ammonium tetrafluoroborate ; 1-Hexyl-3-methyl-imidazolium-tetrafluoroborate ([HMIM]BF₄); 3-glycidoxypropyltrimethoxysilane (KH-570) ; Havoline® XLC (commercial antifreeze coolant)

The present review proposes a comparative analysis of the TC performance of the nanofluids through the introduced new parameter $\mathcal{P}_{TC}$. The first aspect analyzed is the TC enhancements at 30°C, TCE_{30} , for all the GNFs, and the corresponding $\mathcal{P}_{TC}$ (Figure 4). Figure 4a represents TCE_{30} of GNFs in percentage, reported from 2010 until today. For sake of clarity, each year is represented by a different color. High TC enhancements were in 2010 and 2011, with the higher top 3 values (in the 55 – 90 % range) among them. Such TC enhancement values for GNFs were not hitherto reached in the community. A sudden decrease in TCE_{30} (below 20 %) is observable in 2012 and 2013. Since 2014, a little recover in TC enhancement is evident with TCE_{30} found in the 20 – 30 % range. Such TC enhancements are quite good values even if they are lower than the 55 – 90 % enhancement range for the first studies. This range of TC enhancements is also observed until the most recent years.

Figure 4 a) TC enhancement at 30°C for GNF (TCe_{30}) and b) TC performance parameter (P_{TC}) as a function of the publication year for each study.

As shown in the Figure 4b, TCe_{30} (%) was divided by the GE concentration (wt.%) to obtain P_{TC} (equation 2) so that the TC performances for all the prepared nanofluids can be reported in the same graph to be compared. Obviously, higher quantities of GE lead to higher

TC values in the resulting nanofluid. However, it is interesting to consider to what extent the GE concentration impacts the TC of the corresponding nanofluid.

At first sight, it is noticeable that the values are much more spread in Figure 4b than for the representation in Figure 4a. This spreading effect is even more pronounced in figure 4b because of higher scale amplitude. This effect was expected since the GE concentration range is relatively large from 0.005 to 9.3 wt.% for the extreme values. Contrary to what it is observed in figure 4a, figure 4b shows maximum values of P_{TC} shifted to more the recent years, from 2010 - 2011 to 2016 - 2017. Moreover, by taking into account the amount of GE (figure 4b), the highest values of P_{TC} are among the lowest ones for TC₃₀ in figure 4a. That confirms that the behavior related to the TC performance is mainly governed by the GE amount, especially for the lowest GE concentrations. For the highest P_{TC} values, the maximum reported TC enhancements (not so high however), the GE amount inducing these properties was very low: 6, 10 and 17 % of TC enhancement for 0.005[164], 0.01[123] and 0.02[165] wt.% of GE, respectively leading to P_{TC} of 1200, 1000 and 830, respectively. Apart from these three highest P_{TC} values, TC performances of GNF reported these recent years appears to be among the lowest ones. The reported high TC₃₀ are indeed due to the high GE concentration used; tendency that has changed during 2013 to 2017 and tends to arise again in more recent years.

In the following, this review proceeds to analyze in more detail the influence of the mostly used GE derivative materials and the chemical modification strategy used in nanofluid preparation according to the existing literature. The use of defective GE or additives is believed to be responsible of TC lessening (Section II.4). This is the reason why P_{TC} was examined as a function of the main types of GE and the eventual applied chemical modification (figure 5). Among the types of GE (GNP (including GF), GO, rGO, NDG, GQD,

HFGO and FLG, cf. Table 2), GNP, GO and rGO for which the number of published works is relatively high have been considered for analysis.

Figure 5a shows the previously introduced TC performance parameter $\mathcal{P}_{TC}$ as a function of the TC_{e30} and type of both GE and the chemical modification. In the figure, the GE derivatives are represented by different colors and different symbols are used to distinguish among the type of the chemical modification applied.

Even if the data reported in the figure are quite dispersed, regardless of the GE nature, it is clearly noticeable that many points belong to a straight line tendency (gray pointed line was plotted as visual guide, figure 5a). This trend may originate from a sort of standardized behavior applicable to a significant part of the developed nanofluids, about 50 % of them. The nanofluids located on this straight line can be made with GNP, GO or rGO not functionalized, covalently or non-covalently functionalized, the amount of GE required to increase TC of a certain percentage is the same. An augmentation of TC of 10 % results from an addition of 0.1 wt.% of GE within the base fluid for these nanofluids.

Figure 5 a) TC performance parameter P_{TC} as a function of the TC enhancement at 30°C TCe_{30} . b) TC performance parameter P_{TC} as a function of the GE derivative used in the GNF.

There are as well performances of GNFs located below and above this linear trend. The GNFs P_{TC} with values below this particular trend do not show any specificity in the kind of the employed GE nature or the applied chemical modification. It is clear that for these GNF performances a higher GE concentrations were required to induce the reported TC

enhancements, especially for highest TC enhancements up 50-90 % [169–171] since the performance parameter is below 10.

The highest P_{TC} values correspond to quite low measured TCE_{30} , below 20 % with the two highest P_{TC} related to GNFs prepared with non-functionalized GNPs [123,164]. For these works, TCE_{30} is as weak as 6 and 10 %, respectively for the work by Nika *et al.* [123] and Vakili *et al.* [164]. They have indeed used very low GE concentration, *i.e.* 0.01 and 0.005 wt.%, respectively. Interestingly for these GNP-based nanofluids, a very low amount of GNPs has induced remarkable TC enhancements. For the highest P_{TC} reported (1200 %/wt.% of GE), 10 % of TC increase has been induced by only 0.005 wt.% of GNP. That means that the ability of these GNPs to enhance nanofluid TC is increased by a factor of 12 compared to the proportional trend observed. Such performance in GNFs has however seldom been reported. And unfortunately, such high P_{TC} , in the 20 – 50 % improvement range for example, cannot be found in literature. It can be hypothesized that GNPs showing the best TC enhancement can only be dispersed at low concentration in the base fluid.

The P_{TC} of GNP-based nanofluid are relatively high in average compared with that of GO- and rGO-based nanofluids. Obviously, when the researchers would like to improve the nanofluid stability by covalent functionalization by using GO or rGO instead of GNPs, P_{TC} is decreased. And, P_{TC} values from nanofluids prepared with covalent and non-covalent GO/rGO are much lower than those observed for GO and rGO without post-synthesis functionalization, in average. Surprisingly, GO- and rGO-based nanofluids, including after additional functionalization, show higher P_{TC} in average compared with the GNFs prepared with chemically modified GNPs. Functionalized GNPs, which bear a much smaller number of defects than GO (and rGO) would lead to highest P_{TC} even after functionalization.

Another unexpected result concerns the weak difference of P_{TC} of GO- and rGO-based nanofluids. The carbon network recovering of rGO after reduction would increase the intrinsic TC of rGO compared to GO. However, improvement regarding P_{TC} of rGO compared to GO is not obvious (Figure 5b).

Any chemical modification applied to GNPs, *i.e.* addition of surfactant (non-covalent) or introduction/presence of functional groups (covalent), shows a decrease in nanofluid TC. For non-functionalized GNPs, P_{TC} is in the 1000-1200 range (blue square) and the thermal conductivity performance factor is 100 times lower (blue triangle, circle and star) when either a covalent or non-covalent approach is applied. Pristine GNPs are highly hydrophobic and without chemical modification, they can be only dispersed at weak concentration in the base fluid (with high sonication power method). Their chemical modification allows to better disperse them. However, the conducted analysis from Figure 5 shows that when considering P_{TC} , the increase in TC due to the higher content of chemically modified GNPs used is not balanced by their consequent reduction in intrinsic TC due to the structural defects introduced and/or the thermal resistance because of the presence of the surfactant coating. Contrary to GNPs, GO and rGO have already functionalized groups at their surface. As already mentioned, the reduction reaction applied to GO to produce rGO allows a certain recovering of the physical properties, including thermal conductivity (Table 1). This is the reason why chemical modification is less encountered for GO-based nanofluid (or rGO-based nanofluid) preparation. Even if the decrease in thermal conductivity of GO compared to GNP is very significant (Table 2), P_{TC} of GO-based nanofluids is relatively comparable to that of GNP-based nanofluids. The better dispersion expected for GO certainly favorably impacts the nanofluid TC. This finding reveals the complexity and the interplay of the factors affecting TC of graphene-based nanofluids, among which dispersion is an important aspect to consider [174–176]. rGO is expected to show an intermediate (between GNP and GO) behavior since

its transport properties are increased after reduction. Even if, the rate of reduction is difficult to control [47,177], the remaining amount of functional groups helps to disperse them in the base fluids. Like for GO, rGO does not necessarily need any additional chemical modification or high sonication energy dispersion procedures to be used for nanofluid preparation. Moreover, such rGO-based nanofluids have shown interesting P_{TC} (figure 5b). A compromise between the reduction rate and the dispersion ability has certainly to be find for upcoming nanofluid development. When surfactants have been added to improve the dispersion state of rGO in the work from Chen *et al.*, a quite good P_{TC} (450) has been obtained [125].

V Conclusions and outlook

This review covers the development of GNFs and their TC enhancement. The potential of such nanofluids to be used as heat transfer fluids is widely recognized. The different types of GE nanomaterials used for nanofluid preparation are described along with their characteristics and the chemical approaches applied to assist their dispersion in the selected base fluid. The chemical modifications of GE have been reported to potentially strongly influence their intrinsic TC and consequently those of the nanofluid themselves. It appears that many GE systems have been developed and their efficiency is discussed in this review. To this end, a TC performance parameter of GNFs, $\mathcal{P}_{TC}$ is proposed. $\mathcal{P}_{TC}$ allows to relate the TC enhancement of a given nanofluid to the amount of GE used to get this rate of increase. Moreover, this new parameter, never used before, is useful to provide a performance scale and compare the efficiency of the GNFs. The present work allows a rigorous comparative analysis of the GNFs regarding their TC efficiency. The most important findings from this review along with the challenges and future directions are as follows:

V.1 Results

- Over the 10-year of research on GNFs, the highest TC enhancements have been reported in 2010-2011. Since that time, the greater TC enhancements tend to remain stable in the 20 – 30 % range.
- $\mathcal{P}_{TC}$, a universal TC performance parameter, is simply calculated by dividing the TC enhancement by the respective amount of GE required to reach the reported TC. Compared to TC enhancement, the best $\mathcal{P}_{TC}$ are found to be shifted to more recent years, from 2010 - 2011 to 2016 – 2017. In 3-4 recent years, no significant TC performance improvement has been reported in literature.
- A general tendency (followed by around 50 % of the reported studies) showing an augmentation of TC of 10 % with addition of 0.1 wt.% of GE in GNF has been evidenced. The GNFs whose P_{TC} are below this reference have been prepared with a high amount of GE compared to other works. Such conditions may result in GNF viscosity rise which is prejudicial to their further use in circulating systems.
- The best GNFs whose $\mathcal{P}_{TC}$ is situated far above this trend (0.1 wt.% of GE enhancing TC of greater than 10 %) were prepared with the lowest GE concentrations among the published studies.
- Regarding the type of GE, P_{TC} of GNP-based nanofluids are found to be higher in average compared with GNFs prepared with GO or rGO. And $\mathcal{P}_{TC}$ of GNF-based nanofluids is reduced after any chemical modification of GNF by non-covalent or covalent functionalization. P_{TC} for the modified GNPs are much lower than those observed for GO and rGO without post-synthesis functionalization.

V.2 Recommendations

- Study of TC of the GE derivatives themselves (*i.e.* GNP, GF, GO and rGO) used to design GNFs is of great interest to better understand the interdependency between the intrinsic TC of GE at the nanometer scale and the TC measured for the GNFs. Such investigations are poorly found in literature. For example, characterization by scanning transmission electron microscopy (STEM) and electron energy-loss spectroscopy (EELS) usually conducted on CVD grown GE[178] applied to GNP, GO and rGO type would be useful to fill the lack of knowledge currently existing between the nano and the microscale in nanofluid properties.
- More generally, an in-depth characterization to know the physical and structural properties of the GE nanosheets chosen for nanofluid designing would considerably help to go further in the optimization of TC performances. For this purpose, the implementation of advanced characterization methods to study GNF, GO and rGO nanosheets is essential since the structural defects play a significant role in the diffusion of phonons within such nanomaterials. The intrinsic TC of these GE types implying the introduction of functional groups (covalent functionalization) must be investigated. Another related aspect which requires more attention concerns the examination of the interfacial phenomena taking place at the molecular scale when surfactants are adsorbed at the GE surface; in particular regarding phonon scattering at this complex interface and the resulting interface thermal resistance.
- Stability is another aspect to consider when developing nanofluids and especially GNFs. Study of GNF stability is relatively complex since optical techniques such as UV-visible spectroscopy requires dilution of the nanofluid. Moreover, such optical techniques and Zeta potential as well, are carried out to investigate the stability of GNFs under static conditions while most of GNF applications involve fluid flowing.

An important research priority is the investigation of GNF stability under conditions mimicking the envisaged utilizations under heating, flowing and pressure eventually.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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